

Synthesis of freestanding single-crystal perovskite films and heterostructures by etching of sacrificial water-soluble layers

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The ability to create and manipulate materials in two-dimensional (2D) form has repeatedly had transformative impact on science and technology. In parallel with the exfoliation and stacking of intrinsically layered crystals^{1–5}, atomic-scale thin film growth of complex materials has enabled the creation of artificial 2D heterostructures with novel functionality^{6–9} and emergent phenomena, as seen in perovskite heterostructures^{10–12}. However, separation of these layers from the growth substrate has proved challenging, limiting the manipulation capabilities of these heterostructures with respect to exfoliated materials. Here we present a general method to create freestanding perovskite membranes. The key is the epitaxial growth of water-soluble $\text{Sr}_3\text{Al}_2\text{O}_6$ on perovskite substrates, followed by *in situ* growth of films and heterostructures. Millimetre-size single-crystalline membranes are produced by etching the $\text{Sr}_3\text{Al}_2\text{O}_6$ layer in water, providing the opportunity to transfer them to arbitrary substrates and integrate them with heterostructures of semiconductors and layered compounds^{13,14}.

While naturally layered crystals provide intrinsic periodic interfaces for mechanical separation, typically via van der Waals bonding, the generalized synthesis of freestanding membranes from a host carrier essentially requires the creation of an interface to generate bonding anisotropy, to which a force is applied to preferentially break these bonds. The ‘smart-cut’ technique for silicon-on-insulator technology employs an ion-implanted layer confined to a fixed depth from the bulk Si surface, which partially breaks the sp^3 covalent bonding within this layer to form a mechanically cleavable plane¹⁵. As another example, GaN films grown on transparent substrates can be released by absorbing high-power laser irradiation through the substrate, melting the region close to the interface, as used to fabricate substrates for blue-light-emitting diodes¹⁶. Despite their success in electronics applications, a major drawback of these physical release methods is the inevitable structural damage induced, especially in the limit of thin membranes. By contrast, the chemical release method, which involves membrane growth on a sacrificial layer followed by selective etching, is typically less destructive and retains the structural quality of the released membranes. A seminal example is the growth of single-crystal elemental metal membranes on

NaCl crystals, which act as the sacrificial layer¹⁷ released by dissolution in water¹⁸. However, for more complex binary or ternary materials, the selection of both the sacrificial layer and the etchant is restricted by many parameters, such as: etchant selectivity, lattice symmetry and matching for epitaxial growth, and stability of the sacrificial layer during the target membrane growth (often at high temperatures under varying thermodynamic conditions). As a result, there are a limited number of binary or ternary single-crystalline membranes produced through this approach¹⁹. Examples include GaAs membranes released using AlAs as the sacrificial layer²⁰, and SrRuO₃ membranes from SrTiO₃ sacrificial substrates^{21,22}. Both cases utilize the extreme chemical stability of the membrane to the hydrofluoric acid used as the etchant—our aim is to develop a ‘universal’ sacrificial layer and etchant combination, which can be applied to virtually all perovskites and their heterostructures.

To this end, we have developed the epitaxial growth and selective etching of $\text{Sr}_3\text{Al}_2\text{O}_6$ as a hygroscopic oxide sacrificial thin film layer. The isostructural compound $\text{Ca}_3\text{Al}_2\text{O}_6$ is often a few per cent component of Portland cement, forming in the clinker and contributing to binding when hydrated²³. In crystalline form, the hydrogarnet $\text{Ca}_3\text{Al}_2(\text{OH})_{12}$ is only weakly soluble in water due to strong Ca–O bonding. However, by substituting Sr for Ca, the solubility can be greatly increased. As shown in Fig. 1a, $\text{Sr}_3\text{Al}_2\text{O}_6$ forms a cubic unit cell (space group $Pa\bar{3}$) with lattice constant $a = 15.844 \text{ \AA}$, which closely matches four unit cells of the most representative perovskite substrate SrTiO₃ ($a_{\text{STO}} = 3.905 \text{ \AA}$, $4 \times a_{\text{STO}} = 15.620 \text{ \AA}$)²⁴. Despite the apparent structural complexity, with 264 atoms in the unit cell, $\text{Sr}_3\text{Al}_2\text{O}_6$ shares a similar lattice to SrTiO₃, as is clear from the two structures projected onto the (001)-plane (Fig. 1b, only two atomic layers are shown for $\text{Sr}_3\text{Al}_2\text{O}_6$). Compared with a 4×4 SrTiO₃ lattice, 75% of the $\text{Sr}_3\text{Al}_2\text{O}_6$ oxygen sub-lattice is reproduced from SrTiO₃ with small offsets, and the remaining 25% can be regarded as regularly positioned vacancies. Similarly, the $\text{Sr}_3\text{Al}_2\text{O}_6$ cation sub-lattice can be constructed by introducing ordered cation vacancies in the SrTiO₃ cation sub-lattice.

This close resemblance in lattice forms the strong basis for the epitaxial growth of $\text{Sr}_3\text{Al}_2\text{O}_6$ on SrTiO₃. Figure 1d shows the $2\theta - \omega$ X-ray diffraction (XRD) pattern of an 80 nm $\text{Sr}_3\text{Al}_2\text{O}_6$ film grown by pulsed laser deposition on a SrTiO₃ (001) substrate

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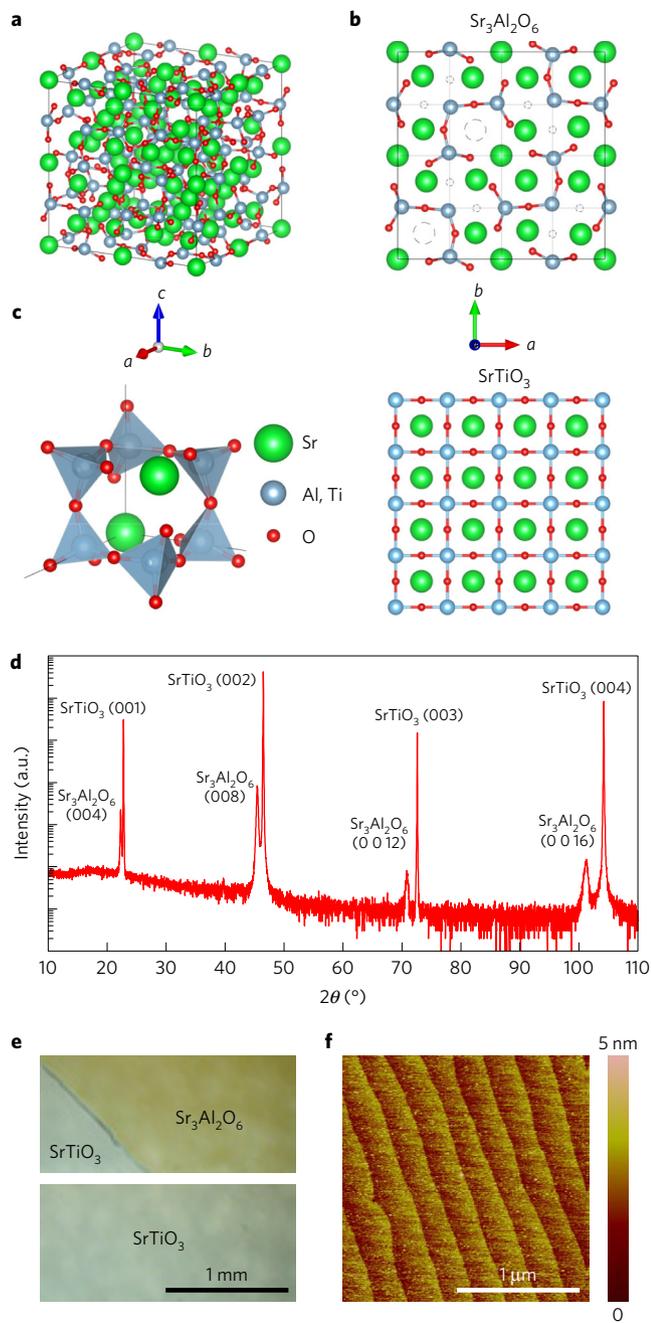


Figure 1 | Crystal structure and epitaxial growth of $\text{Sr}_3\text{Al}_2\text{O}_6$. **a**, Cubic lattice structure of $\text{Sr}_3\text{Al}_2\text{O}_6$. **b**, (Top) Top 1/4 of the $\text{Sr}_3\text{Al}_2\text{O}_6$ unit cell projected onto the (001) plane. Dashed circles indicate vacancy sites. (Bottom) 4×4 unit cells of the SrTiO_3 crystal structure projected onto the (001) plane. **c**, $\text{Al}_6\text{O}_{18}^{18-}$ rings in $\text{Sr}_3\text{Al}_2\text{O}_6$ consisting of AlO_4 tetrahedra. **d**, XRD $2\theta - \omega$ scan of an 80 nm $\text{Sr}_3\text{Al}_2\text{O}_6$ film on a SrTiO_3 (001) substrate. **e**, A masked section of a 200-nm-thick $\text{Sr}_3\text{Al}_2\text{O}_6$ film grown on (001) SrTiO_3 before (top) and after (bottom) water treatment. **f**, AFM image of the surface morphology after dissolving the $\text{Sr}_3\text{Al}_2\text{O}_6$ film by water treatment, showing the recovered SrTiO_3 step-and-terrace structure.

(see Methods), clearly indicating single-phase, epitaxially oriented $\text{Sr}_3\text{Al}_2\text{O}_6$. However, by simply immersing the $\text{Sr}_3\text{Al}_2\text{O}_6/\text{SrTiO}_3$ (001) structure into de-ionized water at room temperature, the $\text{Sr}_3\text{Al}_2\text{O}_6$ layer was completely removed in a few seconds, as shown in Fig. 1e. A key aspect of the Al–O network in $\text{Sr}_3\text{Al}_2\text{O}_6$ is that it consists of discrete 12-membered $\text{Al}_6\text{O}_{18}^{18-}$ rings (~ 10 Å in diameter), composed of six AlO_4 tetrahedra (Fig. 1c). These rings readily

hydrolyse in water, unlike other aluminates with continuous Al–O networks. Therefore, the underlying SrTiO_3 surface is completely free of any residues, recovering the atomically flat SrTiO_3 surface topography characterized by perovskite unit cell steps, as seen by atomic force microscopy (AFM) in Fig. 1f.

The next important question is whether the $\text{Sr}_3\text{Al}_2\text{O}_6$ thin film surface can act as a template for subsequent atomic-scale perovskite growth. Firstly we note indications that the $\text{Sr}_3\text{Al}_2\text{O}_6$ surface preserves the perovskite step-and-terrace structure of the underlying SrTiO_3 substrate, as seen by AFM scans taken immediately after growth (Fig. 2a,b)—the bare surface degrades on the minutes timescale due to atmospheric humidity. Secondly, in this growth regime the $\text{Sr}_3\text{Al}_2\text{O}_6$ film is deposited in the layer-by-layer growth mode, for which the thickness can be controlled by monitoring reflection high-energy electron diffraction (RHEED) oscillations (Fig. 2c). Note that one oscillation corresponds to 1/4 of the $\text{Sr}_3\text{Al}_2\text{O}_6$ unit cell, that is, the perovskite equivalent subunit. For this film, 10 SrTiO_3 unit cells (~ 4 nm) were deposited on top of the $\text{Sr}_3\text{Al}_2\text{O}_6$ film by *in situ* target exchange. High-quality layer-by-layer growth of the SrTiO_3 is evident, as well as a stable atomically flat surface (inset to Fig. 2c). The preservation of the high surface quality can also be seen in the RHEED patterns shown in Fig. 2d,e taken at the end of $\text{Sr}_3\text{Al}_2\text{O}_6$ and SrTiO_3 growth, respectively.

On the basis of these developments, we proceed to discuss the fabrication of large-scale freestanding perovskite membranes. Various perovskite thin films were grown on $\text{Sr}_3\text{Al}_2\text{O}_6$ buffer layers, including SrTiO_3 , $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ (pseudocubic $a_{\text{LSMO}} = 3.868$ Å) and their superlattices (SLs). These heterostructures were immersed in de-ionized water at room temperature to separate the films from their substrates. To mechanically stabilize the films and to facilitate the transfer of the released membranes, an elastomer support layer was formed on the heterostructure surface before release (Fig. 3a; see Methods). Optical microscope images of various millimetre-scale freestanding membranes as released on the polydimethylsiloxane (PDMS) support layers²⁵ are shown in Fig. 3b. By using a commercial screen protector sheet (silicone-coated polyethylene terephthalate, silicone/PET) as the support²⁶, the membranes were successfully released and transferred onto silicon wafers, after which the support layer was fully removed by subsequent heating. Both optical and scanning electron microscope (SEM) images (Fig. 3c) show that the surfaces of the transferred oxide membranes are uniform and intact, even down to few nanometre thicknesses. Further AFM profilometry measurements confirmed that the membrane thickness matches the originally grown thickness to within 2% error, consistent with the residue-free $\text{Sr}_3\text{Al}_2\text{O}_6$ removal by water etching (Fig. 1e,f).

All of the transferred oxide membranes were single crystalline as confirmed by XRD. Representative reciprocal space maps of the SrTiO_3 and $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ films transferred onto PDMS supports around the (103) peak are presented in Fig. 4a–d, showing that the crystalline structure is preserved both before and after the release and transfer processes. The relaxed state of the top SrTiO_3 and $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ films suggests high lattice flexibility of the $\text{Sr}_3\text{Al}_2\text{O}_6$, which renders it responsive to the strain from the top layers, and showing different lattice constants in each case (Fig. 4a,c). A further important issue we wished to address is whether intricate heterostructures could be maintained and released in membrane form, in addition to single-component films. For this the atomic-scale structural coherency of freestanding oxide superlattices was examined by cross-sectional high-angle annular dark-field (HAADF) imaging using scanning transmission electron microscopy (STEM). The HAADF-STEM images of epitaxial and transferred SL samples (with 12 SrTiO_3 (5 unit cell)/ $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ (5 unit cell) repeats) in Fig. 4e,f show that the atomically controlled SL²⁷ preserves its coherent structure after transfer, free from further intermixing or defect formation during the release process.

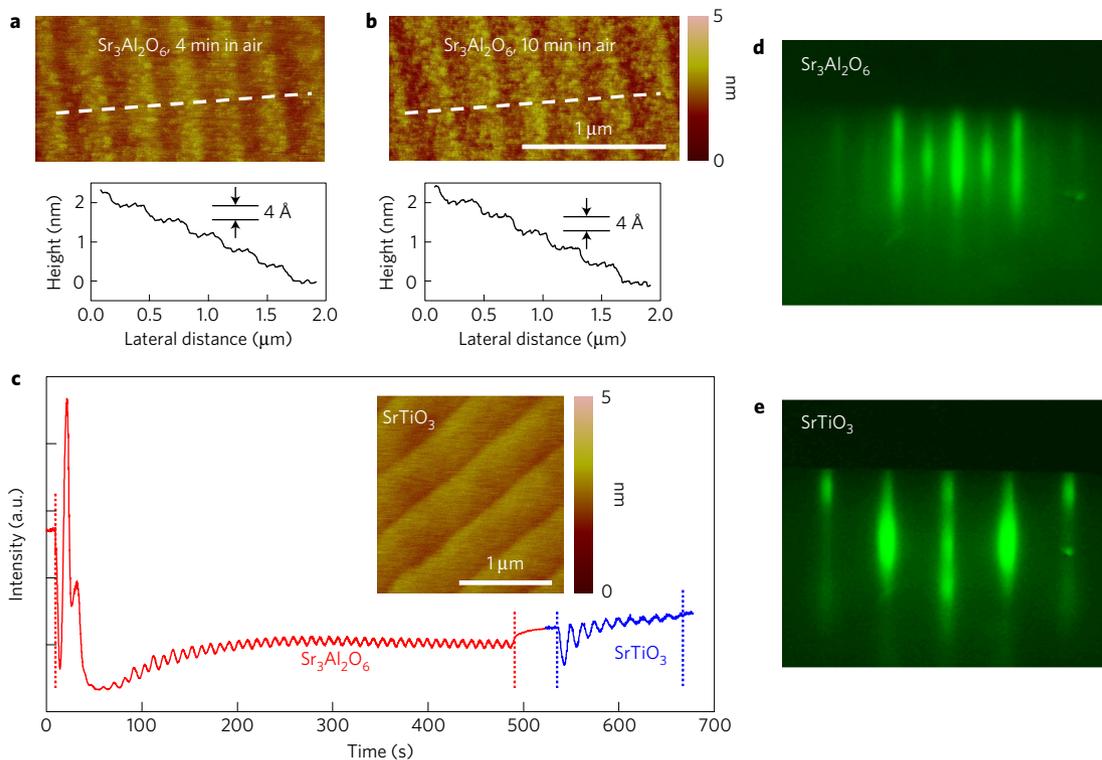


Figure 2 | $\text{Sr}_3\text{Al}_2\text{O}_6$ surface structure. **a, b**, AFM image of the surface morphology of a $\text{Sr}_3\text{Al}_2\text{O}_6$ (20 nm)/ SrTiO_3 (001) sample 4 min (**a**) and 10 min (**b**) after exposure to air. The line profiles indicate that the perovskite unit cell steps of the substrate are preserved at the $\text{Sr}_3\text{Al}_2\text{O}_6$ surface. **c**, RHEED intensity oscillations of a SrTiO_3 (4 nm)/ $\text{Sr}_3\text{Al}_2\text{O}_6$ (20 nm)/ SrTiO_3 (001) heterostructure. 50 oscillations of $\text{Sr}_3\text{Al}_2\text{O}_6$ (red) and 10 oscillations of SrTiO_3 (blue) are shown. Inset: surface morphology of the heterostructure after growth. The vertical dotted lines indicate the start and end of deposition. **d, e**, RHEED diffraction patterns of $\text{Sr}_3\text{Al}_2\text{O}_6$ (20 nm)/ SrTiO_3 (001) (**d**) and SrTiO_3 (4 nm)/ $\text{Sr}_3\text{Al}_2\text{O}_6$ (20 nm)/ SrTiO_3 (001) (**e**).

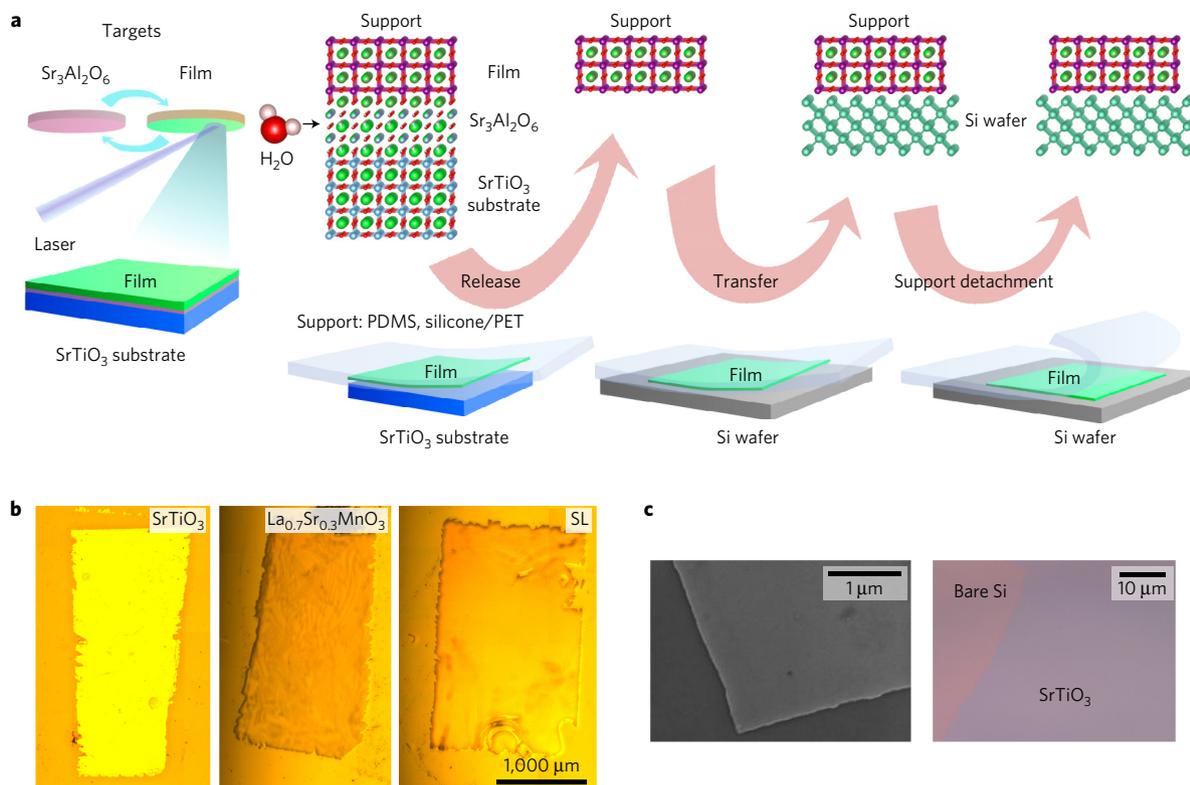


Figure 3 | Synthesis of freestanding perovskite membranes. **a**, Process schematic for heterostructure growth, oxide membrane release and transfer. **b**, Optical microscope images of ~ 80 -nm-thick SrTiO_3 , $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$, and SL (20 SrTiO_3 (5 unit cell)/ $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ (5 unit cell) repeats) freestanding membranes on PDMS support. **c**, SEM image of an 80-nm-thick SrTiO_3 (left) and optical microscope image of a 2-nm-thick SrTiO_3 (right) freestanding membrane transferred onto Si wafers.

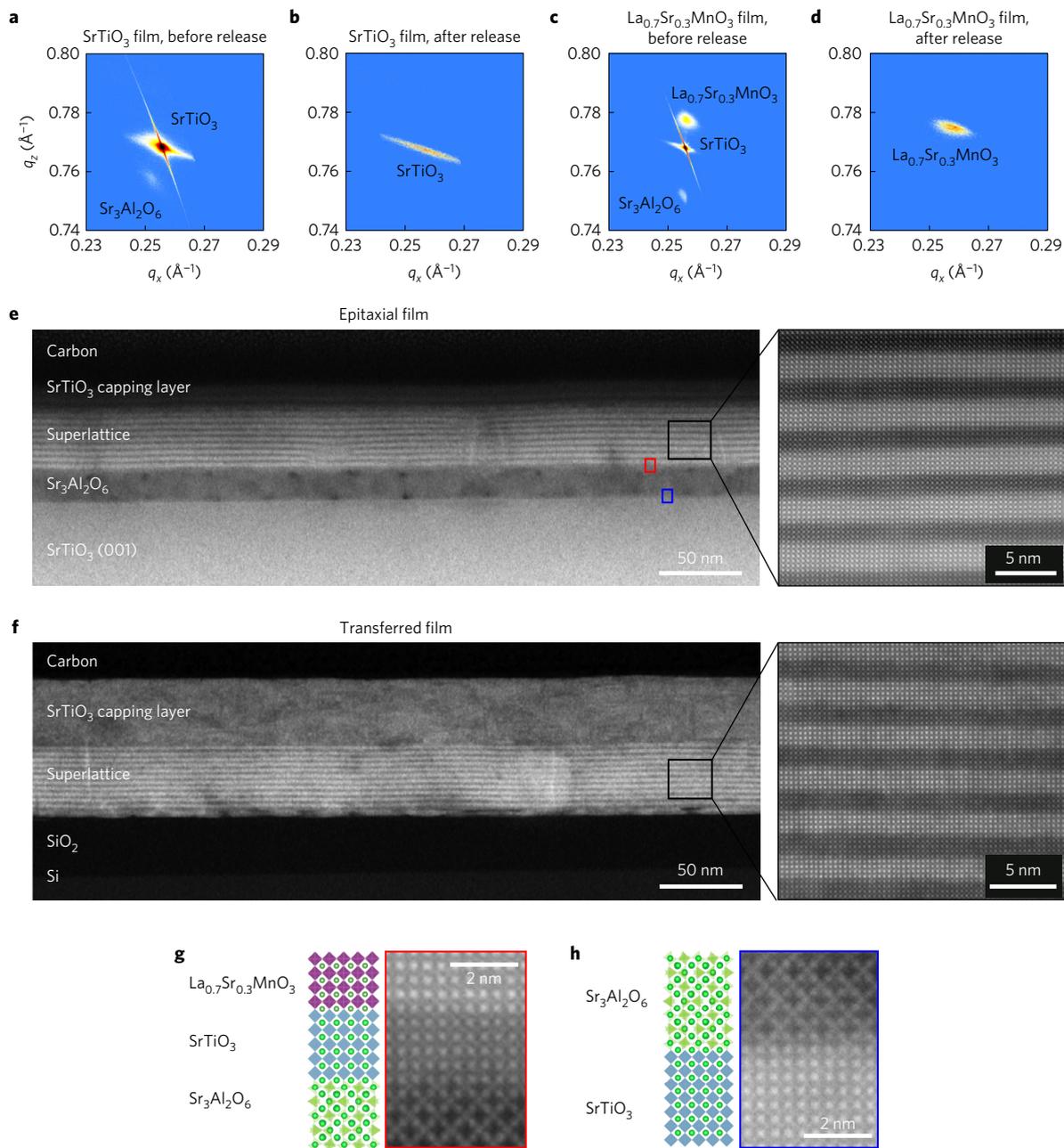


Figure 4 | Crystalline structure of epitaxial membrane films before and after release. **a,b**, Reciprocal space map of a SrTiO₃ (80 nm) film around the (103) peak before (**a**) and after (**b**) release from the substrate. **c,d**, Reciprocal space map of a La_{0.7}Sr_{0.3}MnO₃ (80 nm) film around the (103) peak before (**c**) and after (**d**) release. The difference in the in-plane and out-of-plane lattice constants of Sr₃Al₂O₆ in **a** and **c** arises from the strain effects from the top layers. **e,f**, Cross-sectional HAADF-STEM images of an epitaxial (**e**) and transferred (**f**) La_{0.7}Sr_{0.3}MnO₃/SrTiO₃ SL. Magnified images of the SL region both before and after release are also shown. The brighter layers are La_{0.7}Sr_{0.3}MnO₃ and the darker layers are SrTiO₃, resulting from stronger electron scattering by the elements with higher atomic numbers (La > Sr). **g,h**, Magnified images of the SL/Sr₃Al₂O₆ (**g**) and Sr₃Al₂O₆/SrTiO₃ (001) substrate (**h**) interfaces, together with schematic crystal structures, in reference to the coloured boxes in **e**.

Higher-resolution images of the Sr₃Al₂O₆/SrTiO₃ (001) and the SL/Sr₃Al₂O₆ interfaces (Fig. 4g,h) show close correspondence to Fig. 1b. The rhombic motifs show the Sr atoms in Sr₃Al₂O₆ projected along the [100] orientation. The surface quality and crystallinity were also examined before and after transfer of SL samples (Supplementary Figs 1 and 2). The arithmetic surface roughness changed from 0.17 nm to 0.29 nm, and the rocking curve full-width-half-maximum showed a broadening from 0.07° to 0.20° before and after the film transfer process. We note that contributions to this broadening probably arise from macroscopic distortions of the freestanding superlattice (on PDMS support). The local

surface flatness is comparable to those observed typically in oxide nanosheets³, but over millimetre scales even for this complex SL.

In addition to the structural evolution, we examined the transport and magnetic properties of La_{0.7}Sr_{0.3}MnO₃ and SL films before and after release. For single-layer La_{0.7}Sr_{0.3}MnO₃ films, the physical properties were preserved, and even enhanced, in terms of the Curie temperature (T_C) and residual resistivity (Fig. 5a,b). This change following release is considerably more dramatic for the SL sample. As seen from Fig. 5c,d, the peak in resistivity, typical for ultrathin SLs^{27,28}, increases by ~40 K. Strikingly, the saturation magnetization at low temperatures is apparently doubled for the

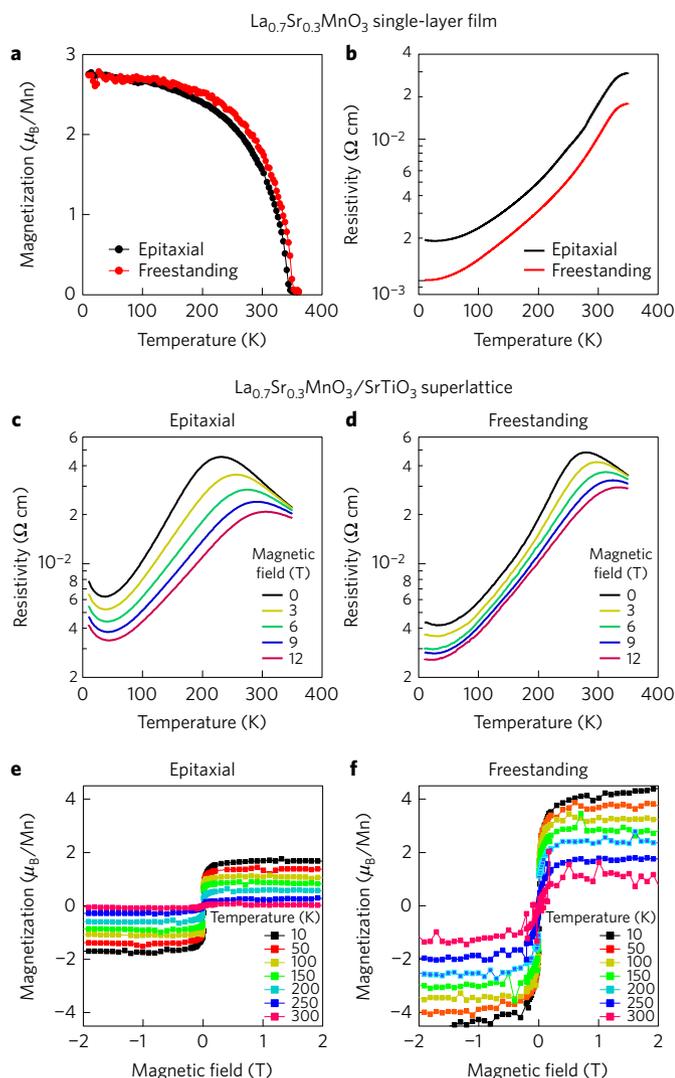


Figure 5 | Magnetic and electrical properties of $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ single-layer and SL films before and after release.

a, Temperature-dependent magnetization of a $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ (20 nm) single-layer film before (black) and after (red) release from the substrate (warming curve measured in 100 Oe in-plane field, after field cooling).

b, Temperature-dependent resistivity of the $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ (20 nm) film before (black) and after (red) release.

c, d, The temperature-dependent resistivity under perpendicular magnetic field for a SL (40 SrTiO₃ (5 unit cell)/ $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ (5 unit cell) repeats) before (**c**) and after (**d**) release from the substrate. **e, f**, In-plane magnetization loops of the SL before (**e**) and after (**f**) release.

freestanding SL (Fig. 5e,f). While the microscopic origin of these phenomena is unclear, the known sensitivity of $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ to strain²⁹, enhanced at interfaces in ultrathin SLs, shows that the strain relaxation during the film release process (Supplementary Fig. 2) can have substantial impact on the physical properties.

These results demonstrate that the use of $\text{Sr}_3\text{Al}_2\text{O}_6$ as an epitaxial sacrificial layer presents an exciting general approach for producing 2D crystalline membranes of perovskite oxides, one of the largest oxide families with a broad range of important physical and chemical properties. The geometry, as well as the large lateral area, is ideal for probing various phenomena including thermal transport, the continuous application of strain, and fabricating oxide electrochemical³⁰ and electromechanical devices³¹. We note that even highly moisture-sensitive perovskites can be incorporated, simply by using ultrathin protective layers in the

membranes. For example, we find that the timescale for dissolving the $\text{Sr}_3\text{Al}_2\text{O}_6$ itself can be varied from ~ 1 s for an unprotected film (Fig. 1e), to ~ 1 day for buried layers of intermediate thickness (20–40 nm), to long-term stability for thinner layers. Considering the atomically controlled growth techniques already well developed in the field, freestanding membranes of oxide single crystals and heterostructures offer a unique opportunity to combine thin film heterostructure approaches with semiconductor device architectures³², flexible electronics³³, and the growing family of exfoliated 2D materials¹⁴.

Methods

Methods and any associated references are available in the [online version of the paper](#).

Received 9 February 2016; accepted 8 August 2016; published online 12 September 2016

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Acknowledgements

This work was supported by the Department of Energy, Office of Basic Energy Sciences, Division of Materials Sciences and Engineering, under contract DE-AC02-76SF00515 (heterostructure synthesis); the Gordon and Betty Moore Foundation's EPiQS Initiative through Grant GBMF4415 (development of release and transfer processes); and the Cornell Center for Materials Research with funding from the NSF MRSEC programme DMR-1120296 (electron microscopy).

Author contributions

D.L. and S.S.H. fabricated and characterized the epitaxial heterostructures and freestanding membranes. D.J.B. and L.E.K. measured and analysed the STEM data. D.L., Y.H. and H.Y.H. designed the experiment and wrote the manuscript, with input from all authors.

Additional information

Supplementary information is available in the [online version of the paper](#). Reprints and permissions information is available online at www.nature.com/reprints. Correspondence and requests for materials should be addressed to Y.H. or H.Y.H.

Competing financial interests

The authors declare no competing financial interests.

Methods

Epitaxial film fabrication. TiO₂-terminated SrTiO₃ (001) substrates were pre-annealed at an oxygen partial pressure (p_{O_2}) of 5×10^{-6} torr for 30 min at 950 °C to achieve sharp step-and-terrace surfaces. The Sr₃Al₂O₆ layer was grown first on top of the SrTiO₃ (001) substrate followed by the growth of thin film (SrTiO₃ or La_{0.7}Sr_{0.3}MnO₃) or their SL by pulsed laser deposition using a 248 nm KrF excimer laser. The Sr₃Al₂O₆ layer was grown at a substrate temperature T_{g} of 700 °C and $p_{\text{O}_2} = 1 \times 10^{-6}$ torr, using 1.25 J cm^{-2} laser fluence and 4 mm^2 laser spot size on the target. The Sr₃Al₂O₆ polycrystalline target was prepared by sintering a mixture of stoichiometric amounts of SrCO₃ and Al₂O₃ at 1,350 °C for 24 h, with two intermediate grinding and pelletizing steps. The Sr₃Al₂O₆ thickness of ~ 20 nm was measured by AFM for 50 RHEED oscillations, indicating that four RHEED oscillations correspond to the deposition of one unit cell of Sr₃Al₂O₆. The SrTiO₃ films and the SL were grown at $T_{\text{g}} = 900$ °C and $p_{\text{O}_2} = 1 \times 10^{-6}$ torr, using 0.35 J cm^{-2} laser fluence and 11 mm^2 laser spot size on a single-crystal SrTiO₃ target²⁷. The La_{0.7}Sr_{0.3}MnO₃ single-layer films were grown at $T_{\text{g}} = 700$ °C and $p_{\text{O}_2} = 1 \times 10^{-5}$ torr, using 0.35 J cm^{-2} laser fluence and 11 mm^2 laser spot size on a polycrystalline target. The entire structure was grown in the layer-by-layer mode as seen from continuous RHEED oscillations for all layers during growth.

Release of freestanding membranes. Approximately 0.1-mm-thick PDMS was spread on a clean silicon wafer to make flat PDMS sheets. The oxide heterostructure surface was adhered onto the PDMS surface. The structure was immersed into room-temperature filtered de-ionized water to dissolve the Sr₃Al₂O₆ layer (Fig. 3a). For transferring the freestanding oxide membranes to other

substrates such as silicon, the samples were attached to commercial screen protector sheets (silicone-coated PET) and released in the same manner. After etching in water, the supports with freestanding membranes were placed on silicon wafers. The membranes remained on the wafers after detaching the supports by heating at 70 °C for 10 min (Fig. 3a).

Characterization. The SEM images were taken using an FEI Nova NanoSEM. The AFM images were acquired in tapping mode. Due to the highly hygroscopic nature of Sr₃Al₂O₆, AFM images of Sr₃Al₂O₆ were taken immediately after the sample growth, with maximum air exposure time of approximately 10 min (Fig. 2b). The XRD data were taken using a monochromated Cu-K _{α} 1 source. Cross-sectional TEM specimens were prepared using the standard focused ion beam (FIB) lift-out process on an FEI Strata 400 STEM FIB equipped with an Omniprobe AutoProbe 200. To study the structure of a freestanding SL film, the film was first transferred onto a silicon wafer as described above. The van der Waals forces between the freestanding film and the silicon substrate were sufficient to subsequently prepare a cross-sectional TEM specimen using the same FIB lift-out process. HAADF-STEM images were obtained on a 200 keV FEI Tecnai F20 SuperTWIN STEM and a fifth-order aberration-corrected 100 keV Nion UltraSTEM. The magnetization data were measured using a superconducting quantum interference device magnetometer with in-plane magnetic field. The freestanding membranes were transferred onto thin PDMS sheets for magnetization measurements. The magnetotransport measurements were conducted in a four-point geometry with gold contacts under perpendicular magnetic field. The freestanding membranes were transferred to polished SrTiO₃ single-crystal substrates as the support template for transport measurements.